

Square-pyramidal subsurface oxygen [Ag₄OAg] drives selective ethene epoxidation on silver

Received: 21 August 2023

Accepted: 27 February 2024

Published online: 25 March 2024

 Check for updates

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Ag-catalysed ethene epoxidation is the only viable route for making ethene oxide (EO) in industry, but the active site remains elusive due to the lack of tools to probe this reaction under high temperature and high-pressure conditions. Here, aided by advanced machine-learning grand canonical global structure exploration and in situ experiments, we identify a unique surface oxide phase, namely O₅ phase, grown on Ag(100) under industrial catalytic conditions. This phase features square-pyramidal subsurface O and strongly adsorbed ethene, which can selectively convert ethene to EO. The other Ag surface facets, although also reconstructing to surface oxide phases, only contain surface O and produce CO₂. The complex in situ surface phases with distinct selectivity contribute to an overall medium (50%) selectivity of Ag catalyst to EO. Our further catalysis experiments with in situ infra-red spectroscopy confirm the theory-predicted infra-red-active C=C vibration of adsorbed ethene on O₅ phase and the microkinetics simulation results.

Ethene epoxidation, a textbook example of kinetically controlled heterogeneous catalysis reactions, holds a central role in the chemical industry, transforming petroleum into valuable chemicals^{1–3}. Silver stands out as the sole catalyst contender for selectively making ethene oxide (EO) while suppressing the combustion to CO₂ (ref. 4). Despite more than 50 years of research, this important reaction remains highly elusive, particularly on those pertinent to the active phase of Ag catalysts and the origin of the EO selectivity occurring at high-pressure industrial conditions^{5–16}. The poor knowledge on the fundamental aspects of the reaction hinders the design of more general epoxidation catalysts for synthesizing alkene (for example, propene) epoxides^{17,18}.

The query about the active site in Ag catalysts dates back to the 1970s^{5,19,20}, but was hampered due to the limitation of surface science techniques, typically operating at a maximum of mbar level that is well below the industrial condition (490–550 K, 2 MPa, with O₂ and C₂H₄

pressures both around 1 bar)⁴. Importantly, the epoxidation selectivity is highly sensitive to O₂ and ethene pressures: under the mbar-level, the EO selectivity is generally below 2% (refs. 21,22), substantially lower than the typical 50% selectivity from unpromoted Ag catalysts under high-pressure conditions (>0.1 bar)²³ and up to 80% through facet engineering by increasing the (100) exposure and reducing stepped sites^{24,25}. The pressure–gap dilemma thus sparks long-time discussions of the reaction mechanism, for example, regarding the origin of the active oxygen species that are transferred to EO²⁶. Via scanning tunnelling microscopy^{9–11,21,27–29} and in situ X-ray photoelectron spectroscopy (XPS)^{14,16,30,31}, the in situ formation of Ag surface oxide phases has been observed, which contain multiple types of oxygen. It was suggested that the surface oxygen species of a 528 eV O_{1s} core-level signal are non-selective to EO^{14,32}, while that with the 530 eV signal, once hypothesized to be the adsorbed O species on metal Ag^{33–36} and recently

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attributed to traces of sulfur (SO_4 units)^{16,37}, was long suspected to be the oxygen source of EO product. There are also other oxygen species, such as subsurface O with O_{1s} signal of -529 eV (refs. 14,15,30), but their catalytic roles are largely unknown¹³. Apparently, no conclusive atomic picture is now available for interpreting all existing experimental observations, which is largely because the active Ag (surface oxides) phases are only present under high-pressure catalytic conditions. The Ag metal and surface oxides identified under low-pressure experimental conditions are both found to be not selective to EO production^{32,38}.

Because of the great complexity of Ag surface structures under reaction conditions, theoretical studies in this field typically utilize bulk-truncated metal or oxide models to gain mechanistic insights^{39–43} and for catalyst screening^{44–46}, while their efforts for revealing Ag phases largely rely on the knowledge from surface science experiments^{47,48}. This is not least because first-principles methods are too computationally demanding to resolve the in situ surface structure even with advanced structure exploration methods⁴⁹. Machine-learning atomistic simulations that emerged in recent years^{50–56}, however, could resolve the long-standing puzzles on ethene epoxidation on Ag catalysts^{57–60}. As demonstrated by us recently, the machine learning-based automated search for optimal surface phases (ASOP) method, a grand canonical global optimization approach, can help to identify the surface phase on Ag surfaces under industrial reaction conditions⁶¹.

In this Article, via theoretical prediction followed by catalysis experimental verification, we identify a thin Ag surface oxide phase as the active phase of ethene epoxidation, which is grown exclusively on Ag(100) under both O_2 and ethene high pressures at 500 K. This is achieved by exploring millions of likely surface oxide structures on major Ag facets and constructing the thermodynamics phase diagram. The identified phase, namely O_5 phase, features the unique subsurface oxygen surrounded by square-pyramidal Ag_3 coordination [Ag_4OAg], which is not present in surface oxides grown on other Ag surface facets ((111), (110)) and also not formed at low ethene or O_2 pressure. We provide theoretical evidence for O_5 phase and confirm its presence by fixed-bed catalytic tests and in situ infra-red (IR) spectroscopy characterization.

Results

Surface structures of Ag catalyst

To determine the relevant Ag phases under reaction conditions, we explore the potential energy surface (PES) of three major Ag facets, that is Ag(111), Ag(100) and Ag(110), under two sets of conditions, that is, the oxidation condition at 500 K and 1 bar of O_2 , and the industrial reaction condition at 500 K and 1 bar of both ethene and O_2 . The ASOP method (Methods and Supplementary Note 1) is used to solve the optimal $\text{Ag}_x\text{O}_y(\text{ethene})_z$ phase for all six tasks (three surfaces at two conditions). In these ASOP simulations, the number of atoms and the supercell size are varied guided by a modified grand canonical Metropolis Monte Carlo scheme, which leads to the exploration of 176 distinct supercells and 11,395 distinct compositions. The ASOP method is based on stochastic surface walking global optimization using global neural network potential^{62–64} (SSW-NN; Methods, Supplementary Tables 1 and 2, and Supplementary Figs. 1–3), as implemented in the large-scale atomic simulation with neural network potential (LASP) code developed by our group⁶⁵. More than three million structure candidates have been explored by SSW-NN. The most stable surface phases at each composition and two conditions are collected and verified using density functional theory (DFT) calculations. All data are detailed in Supplementary Note 2 and Supplementary Figs. 4–7. We found that five tasks result in monolayer Ag surface oxides without ethene adsorption as the optimal phase, and only the task of Ag(100) under the industrial reaction condition results in a distinct phase featuring subsurface O and chemisorbed ethene.

Taking Ag(100) as the example, we show in Fig. 1a the surface energy contour map under the industrial reaction conditions, which

is plotted using the most stable configuration at each $\text{Ag}_x\text{O}_y(\text{ethene})_z$ composition (Ag_xO_y surface oxide at a finite ethene coverage z). The colour represents surface energy (see equation (3) in Methods): the red zone corresponds to the structures with high surface energy, while the blue zone represents structures that are more energetically favourable.

With the most stable phase at each composition, we can plot the surface phase diagram on Ag(100) under various O_2 and ethene pressures, as depicted in Fig. 1b. It shows that as the O_2 pressure increases or temperature decreases, a transition from the metallic Ag(100) surfaces to several different surface oxide phases occurs, as depicted in Fig. 1c, illustrating the structure evolution with highlighting the atomic structure of the key phases. At low pressures (for example, <1 mbar) at 500 K, only the O adsorption phase (orange colour in Fig. 1b) and $p(2\sqrt{2} \times \sqrt{2})$ phase (yellow colour) are present, which were reported in surface science experiments (the inset of Fig. 1b, highlighting the typical surface science conditions)⁴⁸. The atomic O atoms in these phases belong to $\text{O}_4\alpha$ and $\text{O}_4\beta$, respectively, highlighted in Fig. 1d. Notably, under typical conditions for catalytic reactions, the (100) surface turns out to be dominated by another two distinct phases, an $\text{Ag}_2\text{O}_5 c(2\sqrt{2} \times 2\sqrt{2})$ phase (green colour), namely, phase 100 in the region of lower pressure but higher temperature, and O_5 phase (blue colour) in the region of higher pressure but lower temperature. These two phases are close in surface energy— O_5 phase is 0.028 J m^{-2} (0.014 eV per surface Ag) lower than the phase 100 at 500 K and 1 bar of both ethene and O_2 (hereafter referred to as the typical industrial condition). The phase diagram confirms the sensitivity of the surface structure to external conditions. We further elaborate the two critical phases in the following.

Phase 100 holds a $(2\sqrt{2} \times 2\sqrt{2})$ supercell of Ag(100) and has a surface energy of -0.364 J m^{-2} (-0.189 eV per surface Ag), which was recently discovered by us⁶¹. All O species are tetra-coordinated and adsorbed on the surface as $\text{O}_4\alpha$ and $\text{O}_4\beta$ in Fig. 1d.

O_5 phase, represented by an $\text{Ag}_4\text{O}_3(\text{C}_2\text{H}_4)_1$ motif within the smallest $p(\sqrt{5} \times 2)$ supercell (if ethene orientation is considered, the larger supercell with random ethene orientation is slightly more stable, for example, $p(4 \times 4)$ in Fig. 1c), is named for the presence of the unique square-pyramidal subsurface O ($\text{O}_5\beta$). This phase has 1.0 monolayer (ML) Ag and 0.75 ML O coverage. The surface energy is -0.392 J m^{-2} (-0.203 eV per surface Ag, the same below), but if without adsorbed ethene, it will increase to -0.230 J m^{-2} (-0.119 eV). As shown in Fig. 1c with a yellow background, this phase can be decomposed into three layer-by-layer structures, where the $\text{O}_5\beta$ atoms on the (100) facet (at the fourfold hollow site of Ag(100), that is [Ag_4O]) act as the first layer building block, which is then followed by the growth of an Ag–O network that resembles an Ag_2O (111) layer (Supplementary Fig. 5) with a nearly hexagonal Ag–O–Ag skeleton, and finally covered by ethene molecules chemisorbed on Ag atoms that locate at the six-member ring void of Ag–O–Ag skeleton and atop the $\text{O}_5\beta$ atom, that is, [Ag_4OAg]. We emphasize that the square-pyramidal subsurface O has not been reported previously, as O atoms generally prefer a tetrahedral or octahedral geometry, for example, in bulk Ag_2O and on the surface oxides of Ag(111)^{15,30,59}.

Compared with Ag(100), Ag(111) and Ag(110) are much simpler in the phase diagram (Supplementary Figs. 6 and 7), which have the same most stable phase, under both oxidation and industrial reaction conditions. These two most stable phases on (111) and (110) facets are termed as phase 111 and phase 110, respectively. Therefore, under industrial reaction conditions, there are at least four coexisting phases on the Ag catalyst, that is, phase 111 on (111) facet, phase 110 on (110) facet, and phase 100 and O_5 phase on (100) facet. The population of these four phases can be estimated from the Wulff rule and Boltzmann distribution (Supplementary Note 1)—for instance, at 490 K and 1 bar of both ethene and oxygen, they are 0.31, 0.46, 0.02 and 0.21, respectively. The surface energies and atomic structures of phase 111 and phase 110 are briefly summarized as follows.

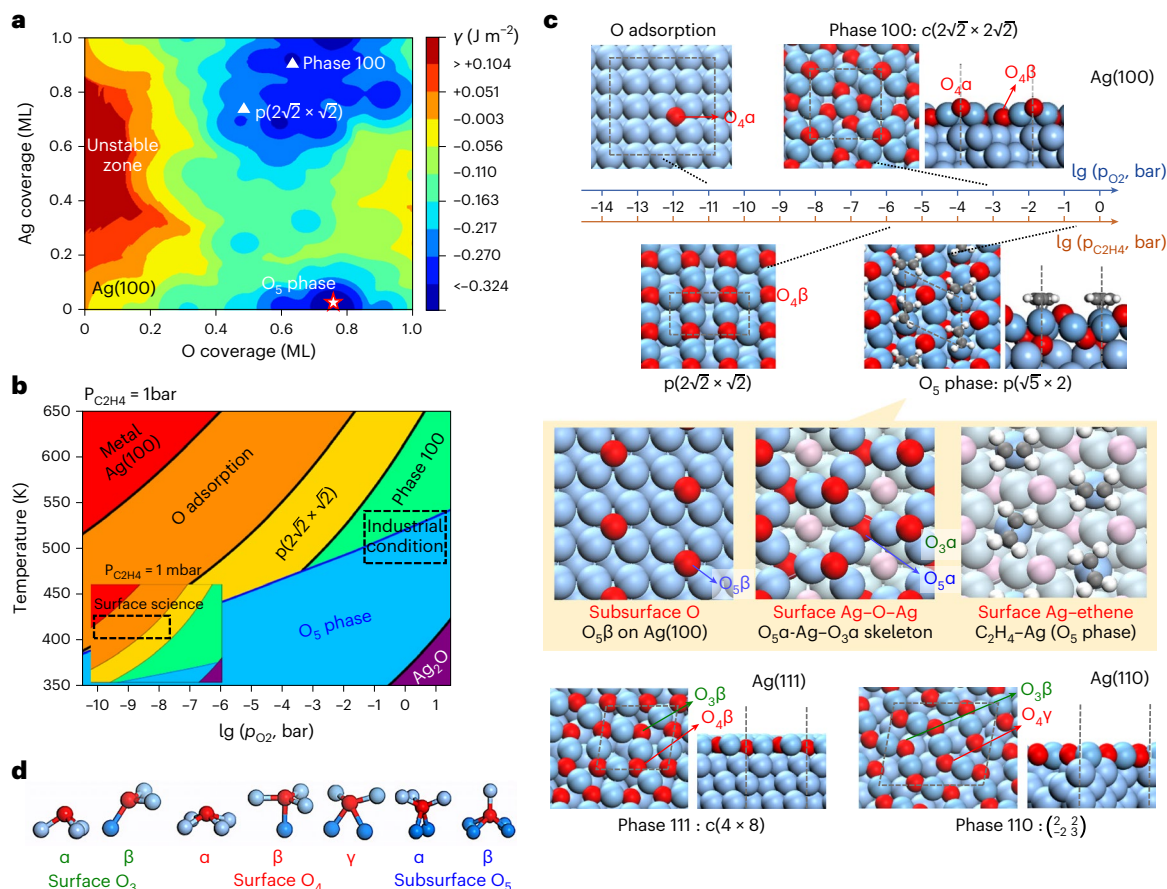


Fig. 1 | Ag surface structure evolution under various conditions. **a**, The potential energy surface contour map of (100) facet at 500 K and 1 bar of both oxygen and ethene from the ASOP simulation. Each point in this map represents an Ag_xO_y(ethene)_z ($z > 0$) composition, where the value of z corresponds to the ethene coverage that yields the lowest surface energy. All the values of coverage are in monolayer (ML). **b**, The surface phase diagram on Ag(100) with the variation of oxygen pressure and temperature, in which ethene pressure is 1 bar for industrial condition (main figure) and 1 mbar for surface science condition (inset). **c**, The diagram of surface evolution on Ag(100) with the changes of

gas pressures at 500 K, and the atomic structures of these phases (top and/or side views). The O₅ surface phase can be viewed as an assembly of three surface structures stacked layer-by-layer, as shown in the yellow background figures (stacking order from left to right). Listed structures (top and side) also include phase 111 and phase 110. **d**, Different types of oxygen species on Ag phases were revealed from ASOP simulations, categorized by the coordination number to surface Ag (blue) and subsurface Ag (deep blue). Colour code includes Ag, blue; C, grey; H, white; and O, red.

Phase 111 is the known $c(4 \times 8)$ reconstruction on Ag(111) and has a surface energy of -0.361 J m^{-2} (-0.162 eV per surface Ag)⁶⁶. All O species on the surface belong to either tri-coordinated (O₃β) or tetra-coordinated (O₄β) oxygen, as shown in Fig. 1d.

Phase 110 has the stoichiometry of Ag₈O₈ in an Ag(110) supercell with the lattice being $u = 2a + 2b$ and $v = -2a + 3b$ ($a = [001]$ and $b = [110]$), consisting of 10 Ag atoms per layer. It has a surface energy of -0.471 J m^{-2} (-0.344 eV per surface Ag). The O atoms on the surface are mostly O₃β atoms, with a minority of O₄γ atoms. Its structure features inter-connected zig-zag [Ag-O-Ag] chains (Fig. 1d), resembling the O-terminated Ag₂O(110) surface.

Electronic properties of the O₅ phase and surface oxides

Now we turn our attention to the differences between O₅ phase and the other surface oxides, which differ in the type of O species and the ethene adsorption ability. To this end, we first calculated the O_{1s} binding energies of all four phases using DFT (Supplementary Table 4), which reveals that the anionic oxygen species in four phases are all in the range of 527–529 eV, consistent with previous XPS experimental findings and theoretical calculations^{12,15,30}. The O_{1s} binding energy of the O₅β atom is in the middle (-528.1 eV).

We note that ethene can readily adsorb on O₅ phase with the calculated adsorption energy of -1.04 eV , which is far larger than its

adsorption on the other surface oxides (approximately -0.4 eV). This is the key reason for the formation of O₅ phase under high pressures of both ethene and oxygen. Figure 2a further compares the projected density of states (p-DOS) peaks of the ethene carbon atoms and the Ag atom involved in adsorption before and after ethene adsorption. It shows that the d -band centre of the Ag atom shifts downward after ethene adsorption, confirming the covalent nature of Ag bonding with ethene. Consistently, a reduction in the intensity of ethene molecular orbitals also occurs after adsorption, including HOMO-1 ($1\pi_g$), HOMO ($2\pi_u$) and LUMO ($2\pi_g$), which suggests the π donation (from ethene to Ag) and back donation (from Ag to ethene) in the ethene-Ag bonding. In addition to π bondings, the HOMO-2 ($2\sigma_g$) orbital of ethene (peak [1] in Fig. 2a) splits into two peaks after adsorption with the major bonding state (at -6.5 eV) indicated by peak [2] in Fig. 2a. The wave function of the state tells that the $2\sigma_g$ of ethene interacts with d_{z^2} of Ag atom, which is activated by the p_z orbital of the underlying O₅β. Moreover, the $2\sigma_g$ of ethene mixes with the p orbital of the surface O₃ atoms, delocalizing to the whole [Ag-O-Ag] oxide network. The electronic structure results confirm the critical role of subsurface O₅β in promoting ethene adsorption, which was long suspected in literatures^{67,68}.

The strong ethene chemisorption behaviour at O₅ phase thus motivated us to explore any measurable quantities for experimental characterization. Figure 2b compares the simulated IR spectra of the

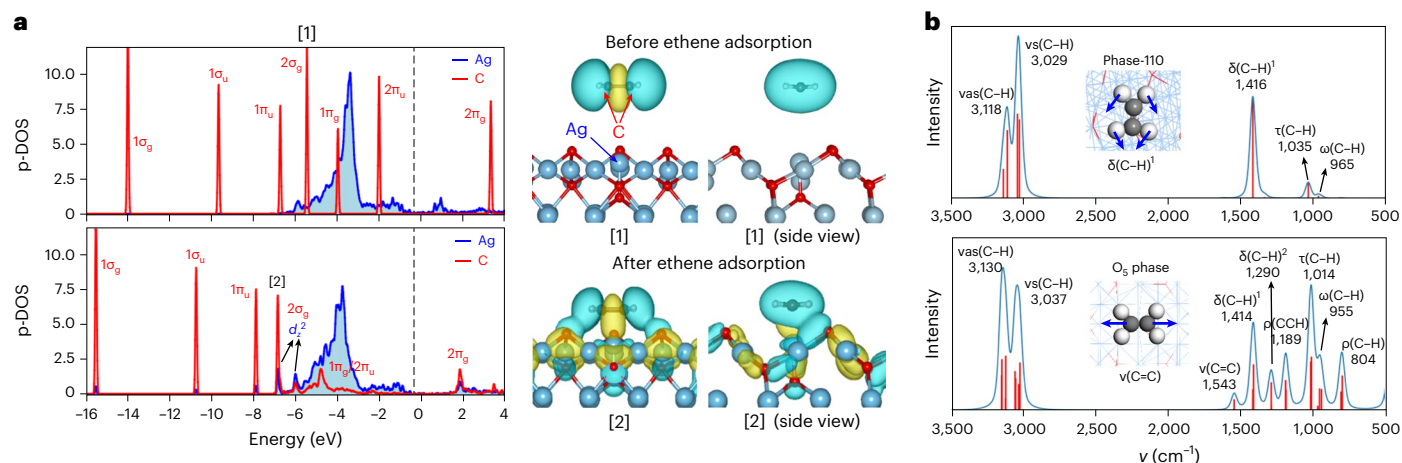


Fig. 2 | Electronic properties and vibrational spectrum of Ag surface oxides.

a, The p-DOS of carbon species (red line) and Ag atom below ethene molecule (blue line) on O_3 phase before (peak [1] referred, ethene is 5 Å above the anchored Ag atom) and after (peak [2] referred) ethene adsorption, in which the dashed line represents the Fermi level. The inset plots the wave functions (negative

in blue and positive in yellow, isosurface value of $2.5 \times 10^{-4} \text{ Bohr}^{-3/2}$) of the $2\sigma_g$ related states, labelled in the p-DOS plot as the peak [1] and [2]. **b**, The simulated vibrational IR spectra of adsorbed ethene on phase 110 and O_3 phase. The colour code is as in Fig. 1.

adsorbed ethene on phase 110 and O_3 phase. For phase 110, the IR active modes correspond to C–H stretching ($\nu(\text{C-H})$) at above $3,000 \text{ cm}^{-1}$, an asymmetric type of C–H scissoring ($\delta(\text{C-H})^1$) at $1,416 \text{ cm}^{-1}$, and the longitudinal bending modes of C–H at about $1,000 \text{ cm}^{-1}$, being the same as ethene in the gas phase. However, on O_3 phase, many more bands are infra-red active. These include the stretching vibration of the C=C bond ($\nu(\text{C=C})$) at $1,543 \text{ cm}^{-1}$, a symmetric type of C–H scissoring ($\delta(\text{C-H})^2$) at $1,290 \text{ cm}^{-1}$, and the rocking modes at $1,189$ and 804 cm^{-1} . Importantly, the $\nu(\text{C=C})$ peak is well known to be IR inactive at $1,640 \text{ cm}^{-1}$ for ethene gas, which becomes IR active after adsorption, redshifting to $1,543 \text{ cm}^{-1}$. This phenomenon can be attributed to the asymmetric Ag–O–Ag chemical environment for the adsorbed ethene interacting with O_3 phase. This observation offers a key clue to verify the presence of O_3 phase in experiments.

Reaction behaviours and overall mechanism

After having determined all phases relevant to industrial conditions, we then explore the reaction pathways on each phase using the automated SSW reaction sampling method based on G-NN potential^{38,69}. The lowest energy pathways on the monolayer surface oxides, including phase 111, phase 100 and phase 110, are detailed in Supplementary Figs. 8–10, revealing that ethene prefers the full oxidation pathway (to CO_2) on these surfaces with phase 110 exhibiting the highest activity. Importantly, only O_3 phase is found to be selective towards EO, and the lowest energy pathway is elaborated in the following.

The DFT-based free energy reaction profile on O_3 phase is shown in Fig. 3a, together with the snapshots of key reaction intermediates shown in Fig. 3b. This lowest energy pathway corresponds to the ethene coverage of 0.125 ML (for surface Ag atoms) with half occupation of $O_3\beta$ sites (Supplementary Note 3). Initially, ethene adsorbs on the Ag– $O_3\beta$ site with the free energy gain by 0.31 eV. Subsequently, the adsorbed ethene combines with a nearby surface O_3 to form oxametallacycle (OMC) intermediate^{7,70–72}, with a barrier of 0.86 eV. Similar to the reaction channels found on metallic Ag surfaces, three potential OMC conversion routes occur via three distinct transition states (TSs), that is, the cyclization to EO, the H-transfer to acetaldehyde (AA), and the dehydrogenation to 2-oxoethyl (OE). Notably, the EO production route possesses the lowest barrier, with its transition state (TS2-EO) being 0.12 eV and 0.14 eV lower in free energy than TS2-OE and TS2-AA, respectively. In TS2-EO, the two [C–O] bonds are not identical in length, being 1.43 Å and 1.85 Å. Finally, the EO molecule, which is only weakly adsorbed on the surface, desorbs into the gas phase (free energy decreases by 0.09 eV).

For the other surfaces, the initial reaction steps are the same, and the major difference lies in the selectivity towards different products, as measured by the height of TS2. Take phase 110, the most active phase for CO_2 production, as an example, where the OMC dehydrogenation route exhibits the lowest barrier, which is 0.56 eV lower than that of the cyclization channel (see the inset table of Fig. 3a).

Once EO/ CO_2 products are formed, the Ag surface is left with O vacancies. The possible pathways for the healing of O vacancies are also explored on these phases (Supplementary Figs. 11–16). For example, the O_2 activation on O_3 phase with O vacancies is found to be kinetically difficult, with a barrier of 1.64 eV for O_2 dissociation and 1.49 eV for the bimolecular reaction ($\text{CH}_2\text{CH}_2 + \text{O}_2$) to form $\text{CH}_2\text{CH}_2\text{OO}$, which is much higher than that of surface O diffusion (0.62 eV barrier). This suggests that O_3 phase alone cannot complete the catalytic cycle of ethene epoxidation. However, O_2 dissociation on the other phases is facile, for example, the barrier of O_2 dissociation with two O vacancies is 0.27 eV and 0.66 eV on phase 111 and phase 110, respectively. Considering that the O diffusion on all these surfaces is low in barrier (<0.78 eV), and phase 111 and phase 110 are another two major phases under reaction conditions, the healing of O vacancies on O_3 phase can be accomplished through the diffusion of O atoms⁷³, for example, from neighbouring phase 111 and phase 110 surfaces. A full microkinetics analysis of all DFT-calculated ethene epoxidation pathways is performed to compare with experimental data in the following section.

We have also examined the stretching frequency of O_2 ($\nu(\text{O-O})$) adsorbed at an O vacancy on phase 111, phase 110, phase 100 and O_3 phase, which are 815, 876, 751 and 744 cm^{-1} , respectively (Supplementary Figs. 12 and 14–16). The wave numbers agree with the in situ Raman spectrum for chemisorbed O–O species, falling within the range of $750\text{--}950 \text{ cm}^{-1}$ (refs. 74–76), indicating the catalyst under reaction conditions should not be a single phase and there is no clear Raman evidence on the presence of O_3 phase in literature.

Catalytic ethene epoxidation experiments

To verify the theoretical predictions on the active phase and the reaction kinetics, we then conducted a series of catalytic experiments in the fixed-bed reactor with fine-tuned reaction conditions. A standard impregnation approach is used to synthesize the Ag/ $\alpha\text{-Al}_2\text{O}_3$ catalyst (Methods). The X-ray diffraction characterization (Supplementary Figs. 17 and 18) of the as-synthesized catalyst confirms the metallic nature of the Ag phase (no detectable Ag_2O peak). The transmission electron microscope (TEM) image in Fig. 4a further reveals the spherical

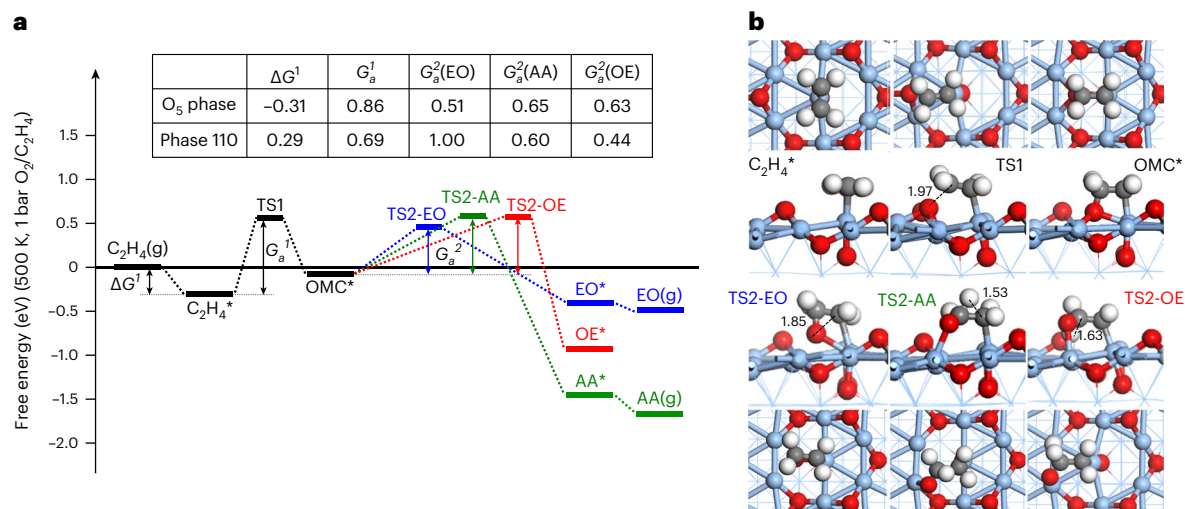


Fig. 3 | Gibbs free energy profile and overall mechanism. **a**, The calculated Gibbs free energy profile of ethene epoxidation on O_5 phase. The elementary barrier and reaction energy (in eV) on O_5 phase and phase 110 are listed in the

inset table. Asterisk indicates adsorbate. **b**, The reaction snapshots along the pathways on O_5 phase, with key bond length (Å) labelled. The colour code is as in Fig. 1.

shape of large Ag nanoparticles, which are supported on $\alpha\text{-Al}_2\text{O}_3$ at a mean particle size of 82 ± 40 nm (standard deviation, the same below). The (200) Ag plane together with the characteristic $\alpha\text{-Al}_2\text{O}_3$ (110) are visible from the high-resolution TEM image in Fig. 4b. The catalytic performance of the synthesized catalyst is collected at a steady state after 1 h of continuous operation. Fig. 4c further demonstrates the good stability of the catalyst in terms of selectivity and conversion performance after more than 200 h of reaction. The catalyst attains a $59.4 \pm 1.2\%$ EO selectivity at a controlled low ethene conversion (1%), exhibiting a slightly elevated selectivity with lower conversion compared with the previously reported experimental data (30–50% selectivity at conversion less than 5%)⁴.

Through systematic variations in reaction temperature and O_2 pressure, we investigate the dependence of EO selectivity on the reaction conditions, as shown in Fig. 4d,e. The EO selectivity is very sensitive to the reaction temperature and achieves the highest value $61.7 \pm 0.5\%$ at 490 K. A change of 50 K in temperature will rapidly reduce the selectivity to ~40%. On the other hand, the EO selectivity is less sensitive to the O_2 pressure, which gradually escalates from $55.8 \pm 0.4\%$ to $62.0 \pm 0.1\%$ with the increase of O_2 pressure from 0.1 to 2.5 bar (Fig. 4e).

Since adsorbed ethene is predicted to exhibit a characteristic C=C stretching mode (Fig. 2b), we carried out in situ Fourier transform IR (FT-IR) spectroscopy and gas chromatography mass spectrometry (GC-MS) to analyse the reaction intermediates and products of ethene epoxidation above 420 K. The existence of both CO_2 and EO in the products is first confirmed by the GC-MS analysis (Supplementary Fig. 19) above 460 K. Figure 4f displays a sequence of FT-IR spectra of the catalyst under 0.1 bar of both C_2H_4 and O_2 with the increase of temperature, where the background spectra collected at each temperature were subtracted (negative/positive peaks, thus indicating the products/the consumed reactants). At temperatures above 500 K, the FT-IR exhibits prominent negative peaks at 2,362 and 2,332 cm^{-1} , which are the symmetric stretching peaks ($\nu_{\text{as}}(\text{OCO})$) of CO_2 . In addition, the negative peaks at 948, 1,338, 1,400, 1,443 and 1,583 cm^{-1} are also observed in FT-IR above 460 K, which can be attributed to C–H longitudinal bending of 948 cm^{-1} , two types of C–H scissoring for 1,338–1,443 cm^{-1} and C=C stretching vibration of 1,583 cm^{-1} (Fig. 2b), respectively. We note that the 1,583 cm^{-1} peak emerges only when both C_2H_4 and O_2 are present under reaction conditions, in companion with the appearance of the CO_2 peak (also see Supplementary Figs. 20 and 21 for time-resolved IR spectra at 500 K). The identification of 1,583 cm^{-1}

peak confirms the theoretical prediction of the IR-active C=C stretching vibration ($\nu(\text{C}=\text{C})$) for adsorbed ethene on O_5 phase.

Apart from in situ IR experiment that confirms the presence of adsorbed ethene, we also conducted in situ isotope (^{18}O) Raman characterization for the Ag/ $\alpha\text{-Al}_2\text{O}_3$ catalyst under the same condition (0.1 bar of C_2H_4 and O_2). As shown in Supplementary Fig. 22a, there are peaks of around 330 cm^{-1} , 610 cm^{-1} and ~750–900 cm^{-1} depending on temperatures, in which the peak at 771 cm^{-1} emerges only after the temperature rises to above 393 K. Supplementary Fig. 22b further indicates that this peak (771 cm^{-1}) disappears during H_2 reduction, but can be retrieved only in a mixed reactant gas ($\text{C}_2\text{H}_4 + \text{O}_2$), characterizing a phase formed only under the reaction conditions. In the subsequent isotope experiment of $^{18}\text{O}_2$, we reveal the red shift of peaks from 887, 771, 333 cm^{-1} to 870, 739, 305 cm^{-1} (Supplementary Fig. 22c), and the formation of E^{18}O (CH^{18}O , m/z of 31) and C^{18}O_2 (m/z of 48) in the online mass spectrum are also detected (Supplementary Fig. 22d). From our DFT calculations, the calculated O–O vibrational frequencies of adsorbed O_2 on O_5 phase and phase 110 are 744 cm^{-1} and 876 cm^{-1} , which red shift to 702 cm^{-1} and 826 cm^{-1} with $^{18}\text{O}_2$ isotope, respectively. This suggests that the experimental phase at 771 cm^{-1} (^{16}O) and 739 cm^{-1} (^{18}O isotope) could correspond to the O_2 adsorption on the O_5 phase that forms under reaction conditions and relates closely with the mixed gas condition.

Microkinetics simulation

Based on the DFT reaction profiles on all major phases, we performed the microkinetics simulation to rationalize the experimental kinetics results. Compared with previous kinetic models^{40,77,78}, our model considers the coexistence of all surface phases on the main Ag facets under reaction conditions. In total, 42 elementary steps on the relevant phases and 42 distinct species are considered in the microkinetics model. All the ordinary differential equations are solved by the backward differentiation formula method⁷⁹, the solution of which is further polished by Newton's method optimization⁸⁰. These are implemented in our recently developed microkinetics-guided machine-learning pathway search package⁵⁵. All the kinetic data of our model are listed in Supplementary Table 7 together with the detailed analysis on the degree of rate/selectivity control, apparent activation energy, reaction orders and the comparison of turnover frequency (TOF) between theory and experiment (Supplementary Tables 8 and 9 and Supplementary Figs. 25–26).

Our microkinetics results are shown in the dashed lines of Fig. 4d,e to compare with the experimental kinetics data. As depicted in the

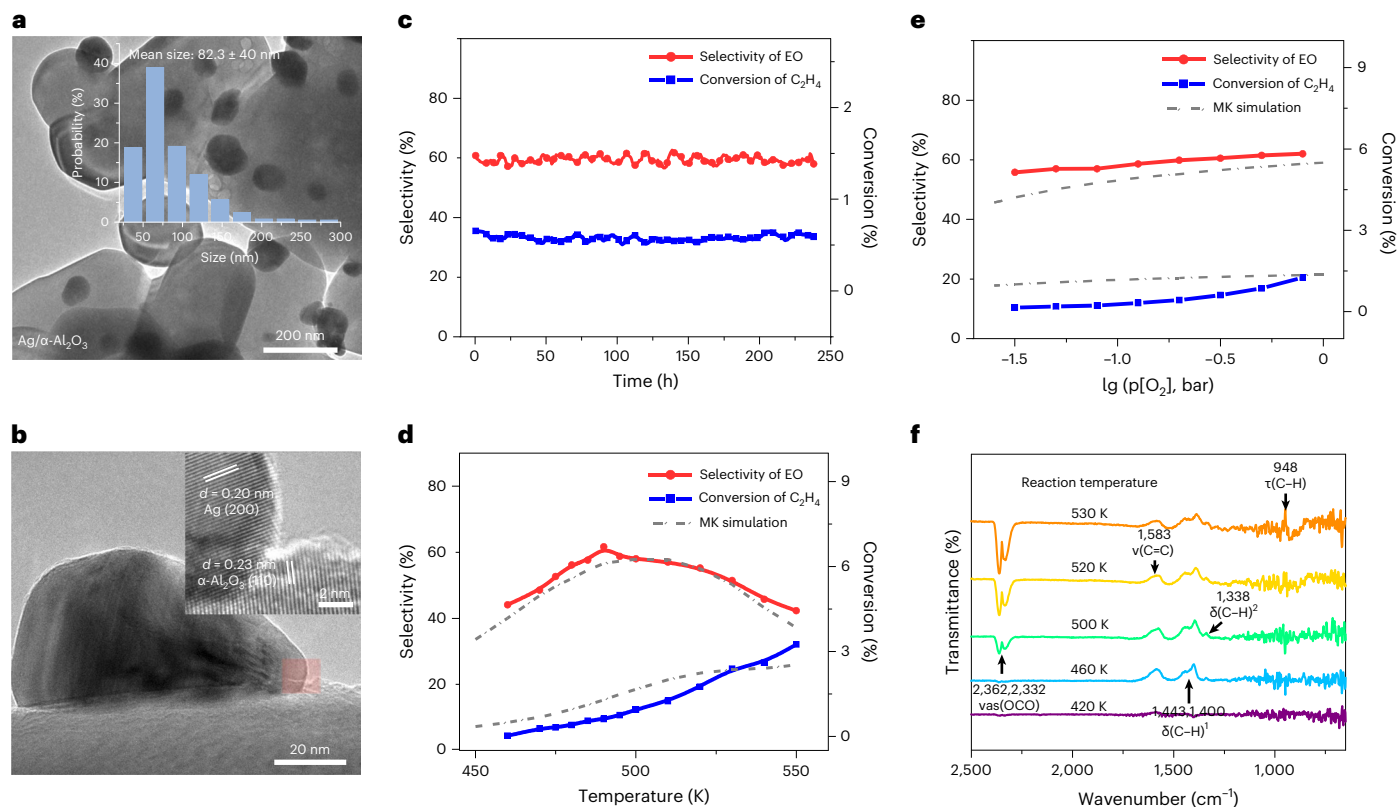


Fig. 4 | Structure characterizations and catalytic performance of Ag/α-Al₂O₃.

a, TEM images and corresponding size distributions of Ag/α-Al₂O₃. **b**, High-resolution TEM image of Ag/α-Al₂O₃. The inset is the magnified image of the rectangular region. **c**, Dependence of C₂H₄ conversion and EO selectivity on time-on-stream for Ag/α-Al₂O₃ at reaction conditions of 490 K, 1 bar C₂H₄, 1 bar O₂ and gas hourly space velocity of 120,000 ml g_{Ag}⁻¹ h⁻¹. **d**, Relationship between the C₂H₄-conversion/EO-selectivity and reaction temperature under 1 bar C₂H₄ and

1 bar O₂ pressure of Ag/α-Al₂O₃ and the simulated results from the microkinetics (MK) model. **e**, EO selectivity and C₂H₄ conversion with increasing O₂ pressure at reaction conditions of 490 K, 1 bar C₂H₄ and the simulated results from the microkinetics model. The range and standard deviation of all the measured selectivity and conversion data are detailed in Supplementary Table 6. **f**, In situ FT-IR spectra as a function of temperature under 0.1 bar C₂H₄ and 0.1 bar O₂ pressure for Ag/α-Al₂O₃.

Table 1 | Reaction rates from microkinetic simulation

Phases	Phase 111	Phase 110	Phase 100	O _s phase
Facet	(111)	(110)	(100)	(100)
Population	0.31	0.46	0.02	0.21
<i>r</i> (EO)	3.7	0.0	0.1	839.7
<i>r</i> (CO ₂)	164.0	1,010.3	7.0	109.7

The microkinetics formation rates to EO and CO₂ (molecule s⁻¹) for the four major phases under reaction conditions. The Ag surface facets on which these four phases grow and the populations of them are also listed. The reaction condition is at 490 K with C₂H₄ and O₂ pressures at both 1 bar.

figures, a good agreement on the trend of kinetics is achieved, which indicates the mechanism and kinetics results from DFT can well capture the impact of temperature and O₂ pressure on selectivity. By analysing the kinetics results, we found that the phase 110 and O_s phase are critical to the overall selectivity—the former favouring CO₂ production and the latter leading to EO products, as reflected by Table 1, where the phase populations and the EO and CO₂ formation rates for all four phases under the reaction conditions are listed.

From Table 1, we can conclude that it is the population of phase 110 and O_s phase grown on (110) and (100) surface facets, respectively, govern largely the EO selectivity. In particular, the population of O_s phase (*w*_{O_s}) is influenced by temperature and the pressure of C₂H₄ and O₂, since it can transform to the non-selective phase 100. Conversely, the population of phase 110 (*w*₁₁₀) can be treated as a constant under

reaction conditions, since it is the only dominant phase on the (110) facet that forms at as low as 10⁻⁸ bar of O₂ pressure (*p*_{O₂}), and the morphology of Ag catalyst (large nanoparticle) is little affected by the subtle changes of the reaction conditions. With these understandings, a simplified kinetics model for computing selectivity *S* is then derived as shown in equation (1).

$$S = \frac{w_{O_s}(P_{C_2H_4} \cdot P_{O_2}, T) r_{O_s}(T)}{w_{110} r_{110}(P_{C_2H_4}, T) + w_{O_s}(P_{C_2H_4} \cdot P_{O_2}, T) r_{O_s}(T)} \quad (1)$$

$$\frac{1}{S} - 1 = \frac{w_{110}}{w_{O_s}(P_{C_2H_4} \cdot P_{O_2}, T) \times \frac{r_{O_s}(T)}{r_{110}(P_{C_2H_4}, T)}}$$

where *w* represents the phase population, *r*_{O_s} is the rate to produce EO, controlled by the surface reaction of ethene, and *r*₁₁₀ is the reaction rate to produce CO₂, which is related to both ethene adsorption step and subsequent surface reactions (Fig. 3a). The equation (1) can be further derived (Supplementary Note 5) to

$$\frac{1}{S} - 1 = e^{(\alpha_1 + \beta_1 T^{-1})} p_{O_2}^{-0.5} + e^{(\alpha_2 + \beta_2 T^{-1})} \quad (2)$$

where α_1 , β_1 , α_2 and β_2 are four parameters determined by the kinetics data— α_2 and β_2 relating to the free energy barriers of EO and CO₂ formation on phase 110 and O_s phase, while α_1 and β_1 further include surface free energy difference between phase 100 and O_s phase. By fitting to our microkinetics results, we have calculated the value of α_1 , β_1 , α_2 and β_2 to be 18, -9.7×10^3 , -14 and 6.8×10^3 , respectively, which

follows the same trend as the value fitted to experimental data, that is, 30, -1.6×10^4 , -8.3 and 3.9×10^3 .

Equation (2) provides a straightforward way to understand the epoxidation kinetics. The volcano-type curve of the selectivity-temperature curve in Fig. 4d is due to the negative value of $\beta_1\beta_2$ (<0), caused by the opposite response of w_{O_5} and r_{O_5}/r_{110} to temperature (the high temperature hinders ethene adsorption, thus lowering w_{O_5} but increasing r_{O_5}/r_{110} , also see Supplementary Note 5). Moreover, the monotonous selectivity- p_{O_2} relation in Fig. 4e is due to $S \propto p_{O_2}^{0.5}$, as p_{O_2} affects positively w_{O_5} by increasing O_5 phase population (see phase diagram in Fig. 1).

The active phase structure revealed here facilitates us to understand why silver is not applicable for longer alkene epoxidation. Our calculations show that propene oxidation on O_5 phase, as depicted in Supplementary Fig. 32, favours the dehydrogenation of the terminal CH_3 group, the so-called allylic hydrogen stripping route⁸¹. This leads to the generation of a stable allyl radical (CH_2CHCH_2), which subsequently undergoes full oxidation. The OMC (CH_3CHCH_2O) intermediate to epoxide is energetically unfavourable by 0.34 eV compared with the allylic hydrogen stripping route. Apparently, this is because while the C=C bond of propene is activated on O_5 phase, the additional methyl group of propene can also interact favourably with the active surface O_3 atom, facilitating C-H breaking in the methyl group and finally leading to deep oxidation. This observation suggests that a universal epoxidation catalyst should be able to both activate the C=C bond and eliminate any possible C-H activation sites. The finding of materials/dopants to enhance the formation of subsurface O and reduce the activity of surface O species should be the direction for designing better alkene epoxidation catalysts.

Conclusions

In summary, through the large-scale grand canonical global structure exploration using G-NN potentials, we uncover a critical surface oxide phase, O_5 phase, from more than three million structure candidates. This phase grows on Ag(100) and exhibits both square-pyramidal subsurface O and strongly adsorbed ethene with special electronic properties under industrial conditions (for example, 1 bar of both O_2 and C_2H_4 at ~ 500 K), but is not favoured thermodynamically under surface science low-pressure conditions when Ag catalyst fails to produce EO with high selectivity. With the structures and the reaction pathways fully explored on all major Ag facets, we show that O_5 phase is the only phase selectively producing EO, in which the EO formation barrier is 0.12 eV lower than that of the combustion routes, while the other surface oxide phases lead to mainly ethene combustion. Importantly, O_5 phase has a low activity towards O_2 dissociation and the O supply of O_5 phase requires the cooperation of the inter-phase and/or inter-termination sites. Guided by theory, our catalysis experiments of home-made Ag/ α - Al_2O_3 catalyst achieve a long-term (>200 h) high EO selectivity ($59.4 \pm 1.2\%$) at 490 K and 1 bar of both O_2 and C_2H_4 . Importantly, the existence of adsorbed ethene on O_5 phase is verified by FT-IR, showing the IR-active C=C stretching vibration at $1,583\text{ cm}^{-1}$, which is the precursor for ethene oxidation. The measured kinetics in temperature-selectivity and O_2 pressure-selectivity relationships agree with the microkinetics simulation and a simplified kinetics model only including O_5 phase (constituting $\sim 21\%$ surface area of the Ag catalyst) and the non-selective (110) phase ($\sim 47\%$ area of catalyst) is derived to explain the observed ethene epoxidation kinetics on the Ag catalyst. This work provides a state-of-the-art solution to reveal the complex, dynamic catalyst evolution under industrial conditions, which could benefit greatly the catalyst design at the atomic level.

Methods

Machine-learning atomic simulation

All the atomic simulations using the SSW method⁶² for PES sampling based on G-NN potentials^{63,64} were carried out using the

LASP package (www.lasphub.com)⁶⁵. The final dataset for the Ag-C-H-O four-element G-NN potential used in this study consists of 62,912 configurations calculated using highly accurate DFT. Each elementary neural network utilizes 305 power-type structural descriptors to predict atomic energies. The network architecture comprises three hidden layers (305-80-50-50-1), with a total of 124,524 learnable parameters. To depict the complex atomic environment in catalysis, the cut-off radius in G-NN potential is set from 1.00 Å to 7.20 Å, and the power (n) used in power-type structural descriptors is up to 16. The hyperbolic tangent function is utilized as the activation function in all the hidden layers. The G-NN potential is in ASCII (text) format and all parameters are human-readable (accessible from the download link provided). The root mean-square errors for energy and force are 5.381 meV per atom and 0.090 eV \AA^{-1} , respectively. Further details on the SSW-NN method, G-NN training and G-NN benchmarking against DFT can be found in Supplementary Methods.

The ASOP algorithm

The ASOP algorithm⁶¹ is utilized to determine the optimal surface phases under a wide range of possible compositions ($Ag_xO_y(C_2H_4)_z$ on various surface periodicities) in a parallel sampling manner. For completeness, here we briefly outline the algorithm of this method. The ASOP simulation is a parallel composition space exploration method. The composition space is discretized into a series of grids, each representing a possible surface periodicity. These grids contain numerous grid points, where each point corresponds to a specific composition (for example, $Ag_3O_2(C_2H_4)_1$). The simulation begins with the coarsest grid (the smallest cell), focusing on compositions with the lowest costs to efficiently sample favourable configurations. Knowledge is then propagated from grid to grid, directing computational efforts towards potentially favourable compositions in finer grids. This process continues until all designated grids have been thoroughly explored.

For every composition, an initial structure is randomly generated. The SSW-NN method is used for global optimization of the PES, yielding the most stable configuration. This configuration is subsequently subjected to thermodynamics analysis. This analysis evaluates composition preferences, that is, the surface free energy, as defined by equation (3).

$$\gamma = \frac{E_{\text{total}} - E_{\text{slab}} - N_{\text{Ag}}\mu_{\text{Ag}} - N_{\text{O}}\mu_{\text{O}} - N_{C_2H_4}(\mu_{C_2H_4} - G_{C_2H_4,\text{ads}})}{A} \quad (3)$$

In this equation, E_{total} and E_{slab} are the energy of the total structure and the Ag slabs, respectively. N and μ are the number and chemical potential of a given species, respectively (see Supplementary Note 1 for the estimation of μ). A is the surface area. $G_{C_2H_4,\text{ads}}$ represents the thermal effects of the adsorbed ethene, including the adsorbate entropies and enthalpies.

DFT calculations

All low-energy structures and key intermediates along the reaction profile were verified using DFT calculations, which were conducted using the plane-wave VASP package⁸². The electron-ion interaction was represented using the projector-augmented wave pseudopotential⁸³, and the exchange functional was evaluated with generalized gradient approximation of Perdew-Burke-Ernzerhof⁸⁴. A kinetic energy cut-off of 450 eV was used, and the convergence criteria for energy and force were set to 10^{-6} eV and 0.05 eV \AA^{-1} , respectively. The Monkhorst-Pack k -mesh was chosen to be 30 times the reciprocal lattice vectors ($1/30\text{ \AA}^{-1}$). For all calculations, including those involving the G-NN potential, the van der Waals correction was accounted for using the DFT-D3 method with zero damping⁸⁵. This correction is crucial for accurately describing the interactions between Ag and O^{10} , as well as molecular adsorption.

To construct the free energy profile, we incorporated zero-point energy and free energy corrections of gaseous species and adsorbates into all kinetic data, which were estimated from standard thermodynamics equations⁸⁶. Various computational methods were also used and compared (Supplementary Table 5) to ensure the accuracy of the kinetic data. It's important to note that except for the PES contour map in Fig. 1a, all data presented in this study are from DFT calculations, allowing for precise comparisons between our theoretical predictions and experimental results.

The Δ SCF method was used to estimate the O_{1s} binding energy. This method calculates the energy difference between the ground state and a state in which a core hole is introduced¹⁵. The calculated value is then adjusted to account for the experimental XPS O_{1s} signal of bulk Ag_2O , which is 529.2 eV (ref. 87).

Surface models

For the Ag surface structure models, we utilized six slab layers for Ag(110) and four slab layers for both Ag(111) and Ag(100). In these models, the bottom two layers were kept fixed. Except for phase 110, supercell models were used for free energy profile calculations to better isolate the reactant from itself in nearby cells. Specifically, a (4×4) supercell of Ag(111) was used for phase 111, and a (4×4) supercell of Ag(100) was used for phase 100 and O_5 phase. All the surface structures of these four phases and their transition states in ethene epoxidation reaction pathways are provided in Supplementary Data 1 in VASP POSCAR format.

Synthesis of Ag/ α - Al_2O_3 catalysts

The α - Al_2O_3 required high-temperature calcination at 1,800 K for 2 h in a muffle furnace to decrease its specific surface area. By following the procedure reported in the literature⁸⁸, Ag/ α - Al_2O_3 was synthesized using silver oxalate ($Ag_2C_2O_4$) as the silver precursor. The $Ag_2C_2O_4$ was obtained by dissolving 10 g of oxalic acid ($H_2C_2O_4$) and 1 g of $AgNO_3$ in 100 ml of deionized water, followed by stirring for 1 h. Subsequently, 1.0 g of $Ag_2C_2O_4$ was mixed with 0.89 ml of H_2O and 0.89 ml of ethenediamine ($C_2H_8N_2$) solution. The resulting mixture was then impregnated onto 3.2 g of α - Al_2O_3 , freeze-dried in vacuum and calcined at 300 °C in air for 4 h. The resulting Ag/ α - Al_2O_3 catalyst had an approximate Ag loading of 18.2% by weight.

Catalytic testing

All catalytic reactions were performed in a continuous flow fixed-bed microreactor. For comparison, all samples contain 50 mg Ag, mixed with 15 g quartz sand. Before the reaction, the catalysts were pre-treated in pure O_2 at 300 °C for 2 h, and then cooled to the reaction temperature. The products were analysed by an online GC-MS (Agilent 8860-5977B) with HP-PLOT Q capillary column. The activity and selectivity of the catalysts were taken from the steady data that is after 1 h in the stream. The ethene conversion and ethene oxide selectivity (S_{EO}) were calculated by using equations (4) and (5), respectively,

$$\text{Conv} = \frac{1/2CO_{2(\text{out})} + EO_{(\text{out})}}{1/2CO_{2(\text{out})} + EO_{(\text{out})} + C_2H_{4(\text{out})}} \times 100\% \quad (4)$$

$$S_{EO} = \left(\frac{EO_{(\text{out})}}{1/2CO_{2(\text{out})} + EO_{(\text{out})}} \right) \times 100\% \quad (5)$$

where in/out subscripts refer to the concentration measured in the inlet/outlet port.

Data availability

All data generated or analysed during this study are included in this published article (and its supplementary information files). Data are also available from the corresponding author upon request.

Code availability

The software code for LASP and the NN potentials used within the article are available from the corresponding author upon request or on the website: LASP software, <http://www.lasphub.com>; the LASP binary code for the Ag-C-H-O system, <http://www.lasphub.com/supportings/lasp-free.tgz>, and Ag-C-H-O G-NN potential, <http://www.lasphub.com/supportings/AgCHO.pot>.

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Acknowledgements

This work received financial support from the National Science Foundation of China (12188101, 22033003, 91945301, 91745201, 92145302, 22122301 and 92061112), Fundamental Research Funds for the Central Universities (20720220011) and the Tencent Foundation for XPLORER PRIZE.

Author contributions

Z.-P.L. conceived the project and guided the research. D.C. designed and performed the theoretical simulations. L.C. designed and performed the experiments. Q.-C.Z. helped on the experiments. Z.-X.Y. helped on the code development and data analysis for the simulation of infra-red spectrum, C.S. helped on the code development. D.C. and L.C. drafted the manuscript. All authors discussed the manuscript and agreed with the content.

Competing interests

The authors declare no competing interests.

Additional information

Supplementary information The online version contains supplementary material available at <https://doi.org/10.1038/s41929-024-01135-2>.

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Peer review information *Nature Catalysis* thanks Marc-Olivier Coppens and the other, anonymous, reviewer(s) for their contribution to the peer review of this work.

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