# Is Transition Metal Oxide a Must? Moisture-Assisted Oxygen Activation in CO Oxidation on Gold/ $\gamma$ -Alumina<sup>†</sup>

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Although a poor catalyst under dry conditions,  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>-supported gold (Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>) turns out to be a superior CO oxidation catalyst under moisture conditions. In this work, extensive density functional theory calculations have been carried out to investigate the physical origin of the moisture promotion effect. By supporting Au strips on the two most stable  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> surfaces, namely, the (110) and (100) faces, we show that the majority (110) surface is catalytically inert due to the saturation of Al cationic sites by dissociated H<sub>2</sub>O. On the other hand, the minority (100) surface in combination with Au is responsible for CO oxidation activity, where O<sub>2</sub> can adsorb at the Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100) interface with a tilted Au-O-O-Al<sub>5c</sub> configuration. In the presence of coadsorbed H<sub>2</sub>O and CO, the adsorption energy of O<sub>2</sub> reaches to 0.7 eV. We find that H<sub>2</sub>O enables the direct dissociation of the reaction intermediate *cis*-OCOO produced by the bimolecular coupling between CO and O<sub>2</sub>, whereas an extra cis-to-trans rotation of OCOO is required in the absence of H<sub>2</sub>O. In the H<sub>2</sub>O-assisted pathway, no atomic oxygen is produced, and the overall barrier is only 0.25 eV, which is 0.28 eV lower than that without H<sub>2</sub>O. By electronic structure analyses, we suggest that a modest acidity of the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (100) surface contributes to the O<sub>2</sub> adsorption, although Al<sub>2</sub>O<sub>3</sub> lacks the d-states that were shown to be important for O<sub>2</sub> activation.

# 1. Introduction

Oxide-supported gold catalyst has received considerable attention in the past decade because it exhibits high activity in many important catalytic conversions.1-12 The activity of goldbased catalyst was shown to be sensitive to both the morphology of gold and the oxide species. A number of key factors were thus summarized, such as the quantum size effect of the gold particle,<sup>3-6</sup> the local coordination of gold,<sup>7,8</sup> the oxide support effect, 9-11,13-15 and the oxidation state of gold.<sup>2,12,16</sup> On the other hand, it was shown recently that the moisture level in the catalytic system is also crucial to activity. In particular, Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> is a poor catalyst for CO oxidation at low moisture levels, but with 3000 ppm H<sub>2</sub>O, its activity for CO oxidation can increase by 2 orders of magnitude, being similar to that of Au/ TiO<sub>2</sub>.<sup>17</sup> The promotional effect of moisture was also identified on Au/TiO<sub>2</sub>, Au/Fe<sub>2</sub>O<sub>3</sub>, Au/MgO (only at high temperatures), and Au/SiO2.17-19 Since the moisture does not change the catalyst morphology dramatically, it was believed that the catalytic mechanism may be different with and without H<sub>2</sub>O.

Various reaction mechanisms have been proposed to understand the moisture effect, focusing on the role of  $H_2O$  on  $O_2$ activation. On Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>, Costello et al.<sup>20</sup> suggested a mechanism that involves a carbonate-like HCO<sub>3</sub> species, which is formed by CO's reacting with the OH and O atom. The HCO<sub>3</sub> can then decompose to release CO<sub>2</sub>. Daté et al.,<sup>17</sup> on the basis of their experimental results on Au/TiO<sub>2</sub> and Au/Al<sub>2</sub>O<sub>3</sub>, suggested that moisture could promote the dissociation of O<sub>2</sub> to produce atomic O at the interface. On the Au/Fe<sub>2</sub>O<sub>3</sub> catalyst, Daniells et al.<sup>18</sup> suggested oxygen vacancy formation on the oxide, which is created after the removal of OH. The vacancy will be later healed by the adsorption and dissociation of O<sub>2</sub>. By density functional theory (DFT) calculations, Bongiorno et al.<sup>21–23</sup> proposed a reaction path on Au<sub>8</sub>/MgO(100) following the Eley–Rideal mechanism, in which an adsorbed  $O_2$ –H<sub>2</sub>O complex reacts with a gas-phase CO. Gold can extract negative charge from surface defects and give part of that charge further to  $O_2$  to bind and activate it to O–O\*.

Among the Au/oxides systems, the Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> system is perhaps the most special one,<sup>17,24,25</sup> since its activity appears to be the most sensitive to the moisture level. The phenomenon is difficult to rationalize under the current knowledge framework. First,  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> is an irreducible oxide (similar to MgO and SiO<sub>2</sub>), which was traditionally regarded as a poor support for Au. Second,  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> is not a transition metal oxide (TMO), which lacks the d-states. The empty d-states were shown to play important roles in activating O2 for CO oxidation on Au/TiO2 and Au/ZrO2.<sup>14,26</sup> To date, it remains unclear that how oxygen is activated on Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> catalyst, and more intriguingly, how water promotes the reaction. In this work, we thus choose the  $Au/\gamma$ - $Al_2O_3$  system as the model system to probe the catalytic role of moisture. We have utilized extensive DFT simulation to map out the possible reaction channels for CO oxidation on  $Au/\gamma$ -Al<sub>2</sub>O<sub>3</sub>. We show that CO oxidation on  $Au/\gamma$ -Al<sub>2</sub>O<sub>3</sub> is governed by the H<sub>2</sub>O-assisted reaction mechanism. H<sub>2</sub>O promotes the CO oxidation by facilitating the decomposition of a reaction intermediate OCOO without the formation of atomic oxygen.

# 2. Calculation details

All DFT slab calculation was performed using the SIESTA package with numerical atomic orbital basis sets and Troullier–Martins norm-conserving pseudopotentials.<sup>27–29</sup> The generalized gradient approximation method, known as GGA-PBE, was used as the exchange-correlation functional.<sup>30</sup> A double- $\zeta$  plus

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**Figure 1.** Bulk and surface models of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>: (a) defected spinel  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> bulk, (b) (110) surface, (c) (100) surface, (O, red; Al, pink; H, white).

polarization basis set was employed.<sup>29,31</sup> An energy shift of 0.01 eV was used to determine the orbital-confining cutoff radii. The energy cutoff for the real space grid used to represent the density was set as 150 Ry. The basis set superposition error was corrected in computing the adsorption energy  $(E_{ad})$  of molecules.  $E_{ad}(X)$  is defined as $E_{ad}(X) = E_{substrate} + E_x - E_{x/substrate}$ , where E is the DFT total energy; a positive value of  $E_{ad}$  means the energy gain during adsorption. The vacuum separation between slabs was more than 16 Å. The geometry relaxation was carried out using the quasi-Newton Broyden method, and the criterion of relaxation for the maximal force on each coordinate was <0.1 eV/Å. For key reaction steps, a criterion of 0.05 eV/Å was utilized to check for convergence, which shows that the difference in adsorption energy or reaction barrier is within 0.05 eV. The transition states of the reactions were searched using our recently developed constrained-Broyden-minimization method<sup>32</sup> and Broyden-Dimer method<sup>33</sup> and confirmed by vibrational frequency calculation via numerical finite difference method. The above calculation setups of SIESTA have been validated by benchmarking some key adsorption energeies with respect to the results from the method based on plane wave basis set. For example, H<sub>2</sub>O adsorption energy on clean  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100) is calculated to be 0.91 eV, but it is 0.88 eV from plane wave methods.

Experimentally,  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> is an intermediate state transformed from an amorphous or boehmite precursor (boehmite  $\rightarrow \gamma \rightarrow \delta$  $\rightarrow \theta \rightarrow \alpha^{34}$ ).  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> appears when boehmite is heated to about 500 °C.<sup>35–37</sup> Because  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> is a transition phase, its composition by definition is not fixed, which is known to typically contain 21-31% tetragonal Al and 69-79% octahedral Al.<sup>38,39</sup> In  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>, hydroxyl is believed to be present due to the incomplete dehydration. Recently, Sohlberg et al.40 proposed a defected spinel model of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>, and the model is also utilized in this work. The optimized cubic cell is shown in Figure 1a (a b = c = 8.030 Å,  $\alpha = \beta = \gamma = 90^{\circ}$ ). The bulk structure can be visualized as follows. In a spinel magnetite cubic cell, four tetrahedral Fe's [the fractional coordinates 7/8(7/8, 7/8, 7/8), (3/8, 7/8, 7/8), (7/8, 3/8, 7/8), (7/8, 7/8, 3/8)] are replaced by hydrogen atoms that attach to lattice O (i.e., hydroxyl), and the rest Fe are replaced by Al.

The (110) and (100) surfaces have been considered as the possible support for gold since they are the two most stable surfaces of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>.<sup>41-43</sup> The bare (110) and (100) surfaces of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> cleaved from the bulk are shown in Figure 1b, c. On the (110) surface, the Al atoms exposed are all four-coordinated (Al<sub>4c</sub>), and the surface oxygen atoms include both two-coordinated O (O<sub>2c</sub>) and three-coordinated (O<sub>3c</sub>) ones; an obvious trough can be seen between two topmost Al–O rows on the surface.<sup>44</sup> On the (100) surface, all the exposed Al atoms are five-coordinated (Al<sub>5c</sub>), and all the O are O<sub>3c</sub>. In our slab calculations, the (110) surface is modeled by a p(2 × 1) unit cell (16.06 × 11.36 Å) (Figure 1b). The slab of the (110) surface contains six oxygen layers (Al layers are between oxygen layers)



**Figure 2.** (a) (110) surface covered by dissociated water, (b) onelayer Au strip on the (110) surface, and (c) two-layer Au strip supported on the (110) surface (O, red; Al, pink; H, white; Au, yellow).



**Figure 3.** Contour plot of the chemical potential (eV) of the gas phase  $H_2O$  at various temperature and pressure conditions. Regions I and II separate at -0.91 eV and the regions II and III separate at -3.10 eV. The white line in region I indicates the chemical potential at the saturated vapor pressure.

with the bottom four layers fixed at the bulk truncated position. The (100) surface is modeled by a  $p(\sqrt{2} \times \sqrt{2})R45^{\circ}$  unit cell (11.36 × 11.36 Å) (Figure 1c). The slab of (100) surface contains four oxygen layers, with the bottom two layers fixed at the bulk truncated position. The convergence of  $E_{ad}$  with respect to the layer thickness has been checked by adding two extra oxygen layers, which shows that the calculated  $E_{ad}$  of H<sub>2</sub>O differs by no more than 0.02 eV. For the (110) surface, only the  $\Gamma$  point was used to sample the first Brillouin zone, and for the (100) surface the **k**-point mesh utilized was (1 × 2 × 1) according to the Monkhorst–Pack scheme.

# 3. Results

**3.1.** Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(110) System. 3.1.1. Structure of Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(110). We first examined the surface phase of the bulk-truncated (110) surface under moisture condition. DFT calculations showed that H<sub>2</sub>O can adsorb strongly on the surface ( $E_{ad}$  = 1.9 eV), and it can further dissociate into OH and H, which is exothermic by 1.2 eV. In total, 3.1 eV will release to dissociate a H<sub>2</sub>O on the surface with respect to a free H<sub>2</sub>O at 0 K. After H<sub>2</sub>O dissociation, the OH adsorbs at a bridge site between two exposed Al<sub>4c</sub> and the H sits on a lattice O, as shown in Figure 2a.

To determine the surface phase, we plotted the contour plot of the chemical potential of the gas phase H<sub>2</sub>O under different temperature and pressure conditions in Figure 3. The chemical potential is computed with respect to the total energy of free H<sub>2</sub>O at 0 K. Figure 3 indicates that the (110) surface is hydrated when the temperature and pressure falls outside the region (III) where the chemical potential of gas phase H<sub>2</sub>O is lower than -3.1 eV. Under the typical calcination conditions of a catalyst (e.g., 600 K<sup>17,45</sup>), the (110) surface is thus certainly covered with dissociated H<sub>2</sub>O. The strong adherence of H<sub>2</sub>O on (110)

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can greatly reduce the (110) surface energy, which is in accord with the experiment finding that the (110) surface is the majority surface.<sup>46,47</sup>

Next, we added Au onto the oxide surface. Experimentally,  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>-supported Au monomer is usually prepared by deposition—precipitation with HAuCl<sub>4</sub> under neutral or weak alkaline solution at room temperature. The samples will then be calcined at ~600 K.<sup>17,24,45</sup> This formation procedure can be represented by the following formula:

$$Au(OH)_3 + 3HAl_5O_8 \rightarrow Au(III)/(Al_5O_8)_3 + 3H_2O$$
(1)

$$Au(OH)_3 + HAl_5O_8 \rightarrow Au(I)/Al_5O_8 + 2H_2O + 1/2O_2$$
(2)

$$\operatorname{Au(OH)}_{3} + \operatorname{HAl}_{5}O_{8} \rightarrow \operatorname{Au(0)}/\operatorname{HAl}_{5}O_{8} + 3/2\operatorname{H}_{2}O + 3/4O_{2} \quad (3)$$

After we tested the Au monomer, dimer, and trimer species adsorption on various sites of the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(110) surface (the calculation procedure similar to our previous work on ZrO<sub>2</sub><sup>26,48</sup>,, we found that the trough of (110) surfaces are the best sites for anchoring Au, and the cationic Au are generally more stable than the neutral Au under typical calcination temperatures. The cationic Au is more stable because the (110) surface is already covered by dissociated H<sub>2</sub>O, and thus, by removing the surface H, the cation Au can bond with lattice O. Therefore, two different models have been considered for Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(110): namely, one and two-layer Au strips on the oxide. The optimized structures are as shown in Figure 2b and c. Due to the absence of some surface H, these Au strips are overall cationic. In the optimized structures, the Au–Au distance is 2.74–2.87 Å, which is close to the Au bulk lattice constant.

**3.1.2.** Activity for CO Oxidation. On the Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(110) systems, we then explored two possible CO oxidation channels. The first one is the CO reacting with molecular O<sub>2</sub>.<sup>13</sup> For CO, it can well adsorb at the edge of Au ( $E_{ad} = 1.29 \text{ eV}$ ). However, O<sub>2</sub> adsorbs neither on the Al<sub>2</sub>O<sub>3</sub>(110) surface nor at the Au/Al<sub>2</sub>O<sub>3</sub>(110) interface because all Al cations on the (110) surface (after the H<sub>2</sub>O dissociation) are capped and saturated. O<sub>2</sub> can only adsorb physically on Au with the optimized O–Au distance being about 2.7 Å. This indicates that the reaction between CO and O<sub>2</sub> cannot follow the Langmuir–Hinshelwood mechanism and is thus ineffcient.<sup>13</sup>

The second possible channel is the CO reacting with the surface OH. Since there is a number of bridging OH groups on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(110) due to the H<sub>2</sub>O dissociation, we also investigated the reaction path for the adsorbed CO to react with a surface bridging OH. However, the reaction between CO and the surface OH was found to be impossible because the surface OH is rather inert. Our attempts to optimize the product OCOH at the interface always lead to the breaking of OC–OH bond, and the structure decays back to the initial CO and OH. Overall, there is no feasible reaction channel for CO oxidation in the Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(110) model system.

**3.2.** Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100) System. 3.2.1. Structure of Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100). The (100) is the second stable surface of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>, where H<sub>2</sub>O was found to adsorb weakly. The  $E_{ad}$  of H<sub>2</sub>O on bare  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100) was calculated to be 0.91 eV. The H<sub>2</sub>O sits on a Al<sub>5c</sub> with the Al–O bond length being 2.04 Å. The dissociation of H<sub>2</sub>O on (100) is a thermoneutral reaction, and the reaction barrier was calculated to be 0.42 eV. After the



**Figure 4.** Structure of a 2-layer Au strip on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100): (a) side view and (b) local view of the interface.

TABLE 1: Adsorption of Oxygen Molecule on Au/  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100)<sup>*a*</sup>

	$O_2$	$O_2 (+ CO^b)$	$O_2 (+ H_2 O^c)$	$O_2 (+ H_2O + CO)$
E <sub>ad</sub> /eV	0.50	0.67	0.61	0.70
<i>d</i> 0−0/Å	1.35	1.35	1.40	1.41
$spin/(\mu_B)$	0.97	0.96	0.23	0.28
Q (lel)	-0.77	-0.77	-0.99	-0.94

<sup>*a*</sup> Q is the Bader net charge<sup>51,52</sup> on the adsorbed O<sub>2</sub>. For comparison, the  $d_{\rm O-O}$  of free O<sub>2</sub> molecule (triplet) is calculated to be 1.25 Å. <sup>*b*</sup>  $E_{\rm ad}$  of CO is 0.95 eV. <sup>*c*</sup>  $E_{\rm ad}$  of H<sub>2</sub>O is 1.00 eV.

dissociation, the H bonds with a nearby surface  $O_{3c}$ , and the OH sits on the top of  $Al_{5c}$ , where the bond distance between Al–OH is 1.78 Å. By comparing to the H<sub>2</sub>O chemical potential at the gas phase (Figure 3), we found that H<sub>2</sub>O adsorbs on the (100) surface in the region (I), while at typical catalyst calcinations temperatures (e.g., 600 K), the (100) surface can be free of H<sub>2</sub>O. Without extra hydroxyls on the (100) surface, the Au on (100) surface is, thus, most likely to be in a charge neutral state.

On the basis of the facts, we built the supported gold model catalyst as follows. A  $p(\sqrt{2} \times \sqrt{2})R45^{\circ}$  unit cell of (100) surface was chosen to anchor the Au strip. We noticed that the lattice of Au and Al<sub>2</sub>O<sub>3</sub> can match rather well in the [110] direction (the lattice mismatch is within 1.4% with reference to the bulk lattice constant). A two-layer Au strip was then added onto the (100) surface. By optimizing the Au strip with different initial registration with respect to the substrate, we found that the best adsorption structure for the Au strip is as that shown in Figure 4a, where a row of Au in the first layer is directly above the Al<sub>5c</sub> (Au–Al distance  $\sim 2.85$  Å), and the other row of Au in the first layer is at the trough bonding with  $O_{3c}$  (Au–O distance ~2.6 Å). The  $E_{ad}$  of the Au strip on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100) is also not high, 0.35 eV per first layer Au atom. This magnitude appears in common for neutral Au on oxide, where 0.46, 0.15, and 0.2–0.3 eV were calculated for Au on rutile TiO<sub>2</sub>, anatase TiO<sub>2</sub>, and ZrO<sub>2</sub>, respectively.<sup>14,49,50</sup> It might be mentioned that the current two-layer Au structure on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100) mimics those on  $TiO_2$  and  $ZrO_2$ ,<sup>14,50</sup> which have been utilized to analyze the activity of CO oxidation. The results here on Al<sub>2</sub>O<sub>3</sub> can thus be compared systematically with the previous calculations. Experimentally, Goodman and co-workers showed that two-layer Au particles are the most active for CO oxidation on Au/TiO2.3-6

**3.2.2.** Adsorption of Reaction Species. On the Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>-(100) model system, we first evaluated the adsorption of the reaction species (CO, O<sub>2</sub> and H<sub>2</sub>O) and their interaction in the coadsorption configurations. The results are summarized in Table 1. The adsorption of H<sub>2</sub>O at the Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> interface ( $E_{ad} = 1.00 \text{ eV}$ ) is slightly stronger than on bare  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>. Molecular oxygen can adsorb at the Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100) interface with its one end resting on Al<sub>5c</sub> and the other end linking with the Au strip. The  $E_{ad}$  of O<sub>2</sub> is 0.50 eV. The interfacial O<sub>2</sub> is still spin-polarized (0.97  $\mu_B$ ) with a bond length 1.36 Å. Our Bader charge analysis<sup>51,52</sup> shows that the net charge on the adsorbed oxygen is -0.77lel, which is dominantly transferred



**Figure 5.** Snapshots of key states in the CO oxidation with and without the presence of H<sub>2</sub>O. (a) Coadsorption of CO and O<sub>2</sub>, (b) TS of CO reacting with O<sub>2</sub>, (c) *cis*-OCOO intermediate, (d) TS for cis-to-trans rotation of OCOO, (e) *trans*-OCOO, (f) TS for O–O bond breaking in *trans*-OCOO, (g) atomic O at the interface, (h) TS for CO reacting with O atom, (i) *cis*-OCOO coadsorption with a H<sub>2</sub>O, (j) TS for O–O bond breaking of *cis*-OCOO neighboring with a H<sub>2</sub>O, (k) two hydroxyls left at the interface, and (l) TS for CO reacting with O–O and the angle of O–O–Au (∠OOAu) during cis-to-trans rotation are labeled. (O, red; Al, pink; C, gray; H, white; O of O<sub>2</sub>, green).

from the Au strip (overall +0.63lel). We also considered the other adsorption geometry of oxygen, in which oxygen is parallel to the oxide surface with both oxygen atoms bonding with an Al cation. In this geometry,  $O_2$  adsorbs only weakly ( $E_{ad}$ = 0.1 eV). It should be noted that the  $O_2$  adsorption structure on Au/Al<sub>2</sub>O<sub>3</sub> interface is different from that on Au/TiO<sub>2</sub> and Au/ZrO<sub>2</sub> systems,<sup>14,50</sup> where the parallel adsorption geometry is preferred generally.

Consistent with previous calculations,<sup>13,14</sup> CO can adsorb on the Au strip, with the  $E_{ad}$  being 0.95 eV. The coadsorption of CO with O<sub>2</sub> can improve the  $E_{ad}$  of O<sub>2</sub> to 0.67 eV (see Figure 5a). Since there is no change in the bond length, the spin polarization and the net charge of the adsorbed O<sub>2</sub> in the coadsorption system, the increase of the  $E_{ad}$  is believed to be due mainly to an electrostatic interaction.

In the presence of H<sub>2</sub>O, we found that H<sub>2</sub>O can help to increase the  $E_{ad}$  of O<sub>2</sub> by ~0.1 eV, which is apparently caused by the formation of a hydrogen-bonding, HOH-O<sub>2</sub> (1.60 Å). The O<sub>2</sub> bond distance is lengthened to 1.40 Å, and the spin polarization of O<sub>2</sub> is largely quenched in the coadsorbed system. With the nearby H<sub>2</sub>O, O<sub>2</sub> becomes more negatively charged by ~0.22lel, and the extra electrons are not from H<sub>2</sub>O, but still from the Au strip according to Bader charge analysis. It implies that O<sub>2</sub> is further activated due to the H<sub>2</sub>O-induced charge transfer. If both H<sub>2</sub>O and CO are present, the  $E_{ad}$  of O<sub>2</sub> at the interface reaches to 0.70 eV. Considering that the chemical potential of oxygen in the gas phase at 300 K, 1 atm is -0.54 eV with respect to the total energy of free O<sub>2</sub> at 0 K.<sup>53</sup> It indicates that the interfacial O<sub>2</sub> is present under reaction conditions.

**3.2.3.** CO Oxidation in the Absence of  $H_2O$ . We then examined the reaction pathway for CO oxidation in the absence of  $H_2O$ . The snapshots of the optimized reaction intermediates are shown in Figure 5a—h, and the determined reaction profile is plotted in Figure 6. The reaction initiates by CO attacking the upper O of molecular  $O_2$  (TS shown in Figure 5b) and forming a *cis*-OCOO complex (Figure 5c, MS<sub>1</sub>) at the interface. This bimolecular reaction step is very facile with the computed reaction barrier only 0.14 eV and the step is exothermic by 0.49 eV. At the *cis*-OCOO state, the C of CO bonds with Au, and the lower O of  $O_2$  sits on the top of an Al cation. Due to the formation of the new OC-OO bond, the distance between Au



**Figure 6.** Schematic diagram to illustrate the energetic difference in CO oxidation on Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> without (blue solid line) and with H<sub>2</sub>O (red dashed line). The structure of states labeled as (c ~ k) can be found in Figure 5. The calculated values (eV) of  $Ea^1$ ,  $\Delta E^1$ ,  $Ea^2$ ,  $\Delta E^2$  are listed in the inserted table.

and the upper O of  $O_2$  is stretched to 2.8 Å. Next, to produce CO<sub>2</sub>, the *cis*-OCOO needs to rotate to a *trans*-OCOO state (TS shown in Figure 5d). The key feature of the trans structure is that the lower O of O2 bonding with Al also links with Au with the O-Au distance being 2.56 A (Figure 5e,  $MS_2$ ). The calculated reaction barrier of the rotation is 0.22 eV. The internal O-O bond at the *trans*-OCOO state can break to release a  $CO_2$ , which needs to overcome a reaction barrier of 0.53 eV. After the TS (Figure 5f), a CO<sub>2</sub> molecule is produced with an atomic O left at the interface bonding with both Au and Al (Au–O 2.1 Å). To facilitate the close O-Au contact, the Au atom in the first layer protrudes significantly (outside from the strip, Figure 5g), with its neighboring Au-Au bond distance increasing from 2.93 to 3.28 Å on average, which reflects the weakening of Au-Au metallic bonding. This interfacial O atom can react with another coming CO with a reaction barrier of 0.37 eV (TS shown in Figure 5h).

It should be mentioned that the rotation of the OCOO intermediate is an indispensable step toward CO<sub>2</sub> production. This is because the atomic O is very unstable on the bare Al<sub>2</sub>O<sub>3</sub> surface (it is 1.62 eV less stable as compared with 1/2O<sub>2</sub> in the gas phase) and a direct O–O bond breaking at the *cis*-OCOO state is kinetically prohibited because the leaving O is far away from the Au strip. On the other hand, the atomic O produced from the *trans*-OCOO can be much stabilized due to its simultaneous bonding with Au and Al ( $E_{ad}(O) = 0.79 \text{ eV}$  with respect to 1/2O<sub>2</sub>). For the same reason, we also found that the direct dissociation of molecular oxygen is kinetically unlikely at the Au/Al<sub>2</sub>O<sub>3</sub> interface.

3.2.4. CO Oxidation in the Presence of H<sub>2</sub>O. In the presence of H<sub>2</sub>O, the CO oxidation pathway becomes much simpler. The important reaction intermediates in the presence of H2O is shown in Figure 5i-1, and the identified reaction path is overviewed as follows. Being similar to that without  $H_2O$ , the first reaction step-namely,  $CO + O_2 \rightarrow cis$ -OCOO (TS shown in Figure 5i)—is facile with a 0.13 eV reaction barrier (the reaction is exothermic by 0.66 eV). In the presence of  $H_2O$ , the second step is the direct bond dissociation of the internal O-O bond of the cis-OCOO (TS shown in Figure 5j). The reaction barrier of this step is only 0.25 eV. In fact, the key reason that cis-OCOO can readily break its O-O bond in the presence of H<sub>2</sub>O is the proton passing from H<sub>2</sub>O to the lower O of OCOO, which stabilizes the lower O by forming hydroxyl at the interface. When another CO comes, it can react with the hydroxyl directly (TS shown in Figure 51), and the proton transfer occurs coincidentally to yield  $CO_2$  and  $H_2O$ . The reaction barrier of the CO + OH step is only 0.18 eV.

By comparing the reaction pathway with and without H<sub>2</sub>O as shown in Figure 6, we can now provide microscopic insight onto the moisture promotion effect. H<sub>2</sub>O acts as a cocatalyst in the system. Through the H-bonding at the initial state and the proton-transfer during reaction, H<sub>2</sub>O activates the O–O bonds of both adsorbed O<sub>2</sub> and the OCOO intermediate. In the presence of H<sub>2</sub>O, the direct cleavage of O–O bond in *cis*-OCOO becomes possible, which saves one elementary step in the reaction profile. Kinetically, this helps to reduce the overall barrier of OCOO decomposition by 0.28 eV (0.53 – 0.25 = 0.28; see Figure 6). ON the basis of Arrhenius law, the presence of moisture can increase the reaction rate by ~10<sup>5</sup> times at 300 K, if assuming the pre-exponential factors are the same.

### 4. Discussion

**4.1. Comparison with Experimental Results.** Our results show that CO oxidation on  $Au/\gamma$ -Al<sub>2</sub>O<sub>3</sub> follows different reaction pathways with and without H<sub>2</sub>O. The presence of H<sub>2</sub>O can reduce the overall barrier of CO oxidation by 0.28 eV and thus increase the reaction rate by 5 orders of magnitude at 300 K. However, Daté has suggested that the mechanism of CO oxidation on Au/Al<sub>2</sub>O<sub>3</sub> is the same under dry and moisture conditions. Their suggestion is based on the experimental finding that the slope of Arrhenius plots is rather constant when the concentration of moisture increases from 2 to 200 ppm, and the measured apparent activation energies are always ~0.23 eV (22 kJ/mol). It is therefore important to ask whether 2 ppm is a low enough moisture concentration to disallow H<sub>2</sub>O adsorption on Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>.

From Figure 3, we found that the chemical potential of the gas phase H<sub>2</sub>O is lower than -1.00 eV only when the moisture concentration is lower than 0.01 ppm at 300 K. Since the H<sub>2</sub>O adsorption energy is 1.00 eV at the Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100) interface, one can expect that H<sub>2</sub>O is still present on the catalyst at the 2 ppm moisture level under the experimental conditions. Because the reaction rates differ significantly with and without H<sub>2</sub>O, CO oxidation on Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> will always follow the H<sub>2</sub>O-assisted reaction pathway with the moisture level from 2 to 200 ppm. This can rationalize why the similar Arrhenius slopes were observed in experiment. The reaction barrier in the presence of H<sub>2</sub>O calculated here (24 kJ/mol) is also consistent with the apparent activation energies in experiment (22 kJ/mol).

It might also be mentioned that carbonate  $(-CO_3, -HCO_3)$ species are not identified in our reaction channels, although the carbonate-involved mechanism has also been suggested in experiment for CO oxidation on Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>.<sup>20</sup> The presence of carbonate has been observed on Au/TiO<sub>2</sub> catalyst, but no clear evidence of carbonate has been reported on Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> systems. Considering that the formation of carbonate is proposed to involve both atomic O and OH at the interface (CO + O + OH  $\rightarrow$  HCO<sub>3</sub>),<sup>54</sup> such reaction channels may, indeed, be available for CO oxidation on Au/TiO<sub>2</sub>, where the atomic oxygen is present. For Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>, due to the dominance of the H<sub>2</sub>O-assisted reaction channel, only OH species are present in the reaction system, and thus, the possibility to have carbonate species is low.

**4.2.** Origin of the Activity on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>. We are now at a position to address why Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> can exhibit as good catalytic activity as Au/TiO<sub>2</sub> but Au supported on other typical irreducible oxides such as MgO and SiO<sub>2</sub> has low activity. Since O<sub>2</sub> adsorption is a known problem on oxides such as Au/MgO,<sup>15</sup> the key issue here is to answer how O<sub>2</sub> is activated on Au/ $\gamma$ -



**Figure 7.** Charge-density-difference contour plots showing the bonding of O<sub>2</sub> at the Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100) interface. Parts a and b are the plots cutting through O<sub>2</sub>-Al and O<sub>2</sub>-Au bonding planes, respectively. The charge density difference is constructed by subtracting the total electron densities of the O<sub>2</sub> adsorbed model O<sub>2</sub>/Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100) from an O<sub>2</sub> and a Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> slab, each in the same structure. The unit of charge density is e/Å<sup>3</sup>.

Al<sub>2</sub>O<sub>3</sub>. In fact, we have shown from DFT that O<sub>2</sub> adsorption geometry on Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> is different from that on Au/TiO<sub>2</sub> and Au/ZrO<sub>2</sub>. At the Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> interface, O<sub>2</sub> prefers the one-end Au-O-O-Al<sub>5c</sub> adsorption geometry (Figure 5a). By contrast, at the Au/TMO, O<sub>2</sub> prefers the parallel adsorption geometry, with both O atoms of O<sub>2</sub> bonding with a TM cation.

To understand further how O<sub>2</sub> interacts with Al<sub>2</sub>O<sub>3</sub> and Au, we have analyzed the electronic structure the system before and after the O<sub>2</sub> adsorption. Figure 7 shows the contour plots of the electronic charge density difference, constructed by subtracting the total electron densities of the O<sub>2</sub> adsorbed model (O<sub>2</sub>/Au/  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>) from the separated O<sub>2</sub> and Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> slab. The contour planes are through the bonding plans of O2 with Al and Au, respectively. Figure 7a shows that only polarization of oxygen atom can be seen at the  $O_2/Al$  bonding plane, and no covalent bonding (orbital mixing) is found between O and Al. This indicates that the interaction between oxygen and the cationic site of the oxide support is purely electrostatic interaction. On the other hand, the  $2\pi^*$  of  $O_2$  can covalently bond with the d orbital of gold, as shown in Figure 7b, where the d orbital of gold toward  $O_2$  depletes while the  $2\pi^*$  of  $O_2$ accumulates electron. The bonding picture of O<sub>2</sub> with oxide (Figure 7a) is distinct from that of O<sub>2</sub> adsorption on Au/TMO.<sup>14</sup> For O<sub>2</sub> adsorption on Au/TMO, obvious  $2\pi^*(O_2)-d(TM)$ bonding can be seen between  $O_2$  and the oxide.

On the basis of electronic structure analyses, we can conclude that  $O_2$  adheres to the Al cation via the acid—base electrostatic interaction, and thus, the one-end adsorption geometry is preferred. For comparison, the  $E_{ad}$  of  $O_2$  on Au/Al<sub>2</sub>O<sub>3</sub> interface is not high, being around 0.6 eV. On TMO, by contrast, the  $2\pi^*$  of  $O_2$  forms the covalent bonding with TM cation the d orbital with a three-center  $2\pi^*$ -d bonding. The  $O_2$  prefers to adsorb on TM cation with a parallel geometry. With the covalent bonding, the  $O_2$  adsorption on Au/TMO interface is thus systematically stronger, from 0.8 to 1.7 eV, based on the substrates (TiO<sub>2</sub> and ZrO<sub>2</sub>).<sup>14,50,55</sup>

Finally, we emphasize that the Lewis acidity of the oxide surface controls the coverage of surface hydroxyls and thus should not be too strong for a surface to be catalytically active. Taking the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(110) surface as the example, H<sub>2</sub>O adsorbs dissociatively on the exposed Al<sub>4c</sub> sites. The surface ends up with the termination by hydroxyl groups at the room temperature, which finally deactivates the surface. The strong acidity of Al<sub>4c</sub> on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(110) is clearly not ideal for catalysis. For the same reason, it is conceivable that a too high concentration of moisture in the Au/oxide catalyst may poison CO oxidation,

since  $\rm H_2O$  and  $\rm O_2$  will compete for adsorption at the interfacial cationic sites.  $^{56}$ 

#### 5. Conclusion

This work represents our theoretical attempt to rationalize the CO oxidation activity on Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>, focusing on how moisture promotes the reaction microscopically. The majority (110) and the minority (100) surfaces of  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> are utilized as the substrates for supporting gold. We demonstrate that CO oxidation on Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> occurs at the minority sites—namely, Au on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100)—and is governed by the H<sub>2</sub>O-assisted reaction pathway. The main results are outlined as follows:

(i) The Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(110) system shows no catalytic activity at low temperatures. All the exposed Al sites on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(110) are blocked by dissociated H<sub>2</sub>O due to the high activity of unsaturated Al<sub>4c</sub>. Oxygen adsorbs only physically on gold. CO can adsorb on gold but cannot react with the interfacial hydroxyl.

(ii) On Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100), oxygen can adsorb at the interface with a tilted one-end Au–O–O–Al<sub>5c</sub> configuration. The  $E_{ad}$ of O<sub>2</sub> is 0.7 eV with the coadsorption of CO and H<sub>2</sub>O. H<sub>2</sub>O can also adsorb molecularly at the Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>(100) interface with an  $E_{ad}$  of 1.00 eV, and the dissociation of H<sub>2</sub>O is a reversible thermoneutral reaction.

(iii) In the absence of H<sub>2</sub>O, CO oxidation follows the CO +  $O_2 \rightarrow cis$ -OCOO  $\rightarrow trans$ -OCOO  $\rightarrow$  CO<sub>2</sub> + O and CO + O  $\rightarrow$  CO<sub>2</sub> pathway. Because the atomic O is very unstable on a bare Al site, a cis-to-trans conversion of OCOO is needed to produce CO<sub>2</sub>, and the overall barrier is 0.53 eV.

(iv) In the presence of H<sub>2</sub>O, CO oxidation follows the CO + O<sub>2</sub> + H<sub>2</sub>O  $\rightarrow$  *cis*-OCOO + H<sub>2</sub>O  $\rightarrow$  CO<sub>2</sub> + 2OH and CO + 2OH  $\rightarrow$  CO<sub>2</sub> + H<sub>2</sub>O pathway. There is no atomic O involved in the H<sub>2</sub>O-assisted pathway, and the overall barrier is only 0.25 eV. H<sub>2</sub>O can activate the O–O bond in both the molecular oxygen and the OCOO intermediate via the H-bonding.

(v) The bonding picture for  $O_2$  at the Au/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> is different from that for  $O_2$  at Au/TiO<sub>2</sub> and Au/ZrO<sub>2</sub>. In the Al<sub>2</sub>O<sub>3</sub> system,  $O_2$  interacts with cationic Al via purely electrostatic interaction without the orbital mixing. Despite the lack of d-states in Al<sub>2</sub>O<sub>3</sub>, a modest Lewis acidity of Al cationic sites on (100) surface contributes to the O<sub>2</sub> adsorption and activation.

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